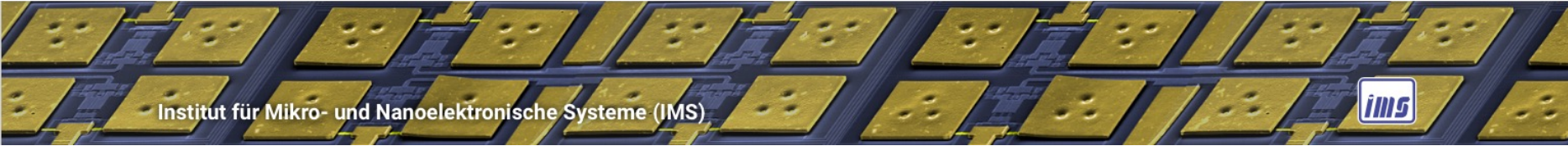
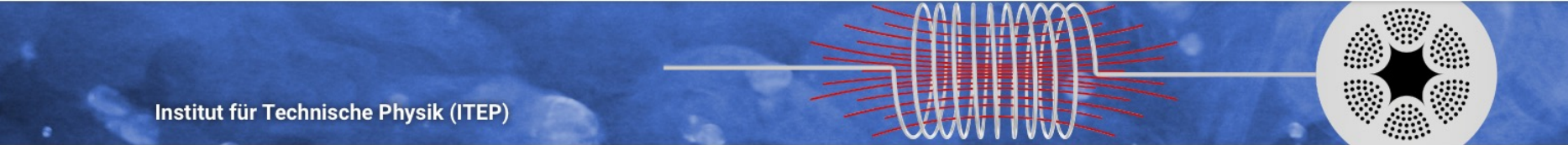


Superconductivity for Engineers

Prof. Dr. Sebastian Kempf, Prof. Dr. Bernhard Holzapfel
Summer term 2021



Institut für Mikro- und Nanoelektronische Systeme (IMS)



Institut für Technische Physik (ITEP)

(Preliminary) Schedule

	Day	Date	Lecture / Tutorial	Day	Date	Lecture / Tutorial
1	Mon	21-04-12	Lecture 1 (SK)	Wed	21-04-14	
2	Mon	21-04-19	Lecture 2 (BH)	Wed	21-04-21	
3	Mon	21-04-26	Lecture 3 (SK)	Wed	21-04-28	Tutorial 1 (IMS)
4	Mon	21-05-03	Lecture 4 (SK)	Wed	21-05-05	
5	Mon	21-05-10	Lecture 5 (SK)	Wed	21-05-12	Tutorial 2 (IMS)
6	Mon	21-05-17	Lecture 6 (SK)	Wed	21-05-19	Tutorial 2 (IMS)
7	Mon	21-05-24	---	Wed	21-05-26	
8	Mon	21-05-31	Lecture 7 (BH)	Wed	21-06-02	Tutorial 3 (IMS)
9	Mon	21-06-07	Lecture 8 (BH)	Wed	21-06-09	Tutorial 4 (ITEP)
10	Mon	21-06-14	Lecture 9 (BH)	Wed	21-06-16	
11	Mon	21-06-21	Lecture 10 (BH)	Wed	21-06-23	Tutorial 5 (ITEP)
12	Mon	21-06-28	Lecture 11 (BH)	Wed	21-06-30	
13	Mon	21-07-05	Lecture 12 (BH)	Wed	21-07-07	Tutorial 6 (ITEP)
14	Mon	21-07-12	Lecture 13 (SK)	Wed	21-07-14	
15	Mon	21-07-19	Lecture 14 (SK)	Wed	21-07-21	Tutorial 7 (IMS, ITEP)

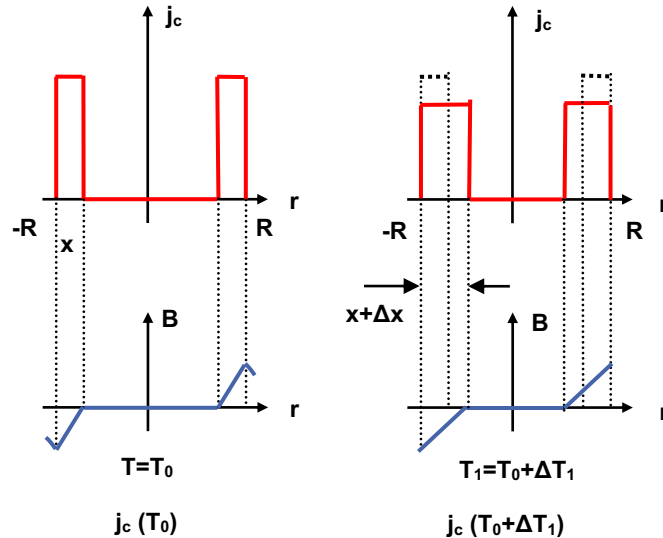
(Preliminary) Lecture content

- Lecture 1: (SK) Introduction and overview
- Lecture 2: (BH) Superconductor applications
- Lecture 3: (SK) Normal metals and properties of the normal conducting state
- Lecture 4: (SK) Perfect conductor, ideal diamagnetism, Two-Fluid-Model, London theory
- Lecture 5: (SK) Disordered superconductors, Pippard theory, microwave properties
- Lecture 6: (SK) BCS theory
- Lecture 7: (BH) Type-II superconductors, Current transport**
- Lecture 8: (BH) Bean Model, Ginzburg-Landau theory**
- Lecture 9: (BH) GL theory, intermediate state**
- Lecture 10: (BH) Superconducting Materials for applications**
- Lecture 11: (BH) Electrical stabilization and thermal aspects**
- Lecture 12: (BH) ac-losses, pinning in HTS**
- Lecture 13: (SK) Josephson junctions and SQUIDs
- Lecture 14: (SK) Josephson junctions and SQUIDs

Stability of the sc state in sc wires

- Thermal balance
- Current sharing
- **Intrinsic filament stability**
- Electric stability with cooling

Requirement for thermal stability



Macroscopic flux penetration depth

$$x = \frac{B}{\mu_0 j_c}$$

Penetration depth change Δx upon temperature rise ΔT_1

$$\Delta x = -\frac{B}{\mu_0 j_c^2} \Delta j_c \quad \text{mit} \quad \Delta j_c = \frac{dj_c}{dT} \Delta T_1$$

Energy dissipation due to flux motion ΔQ

$$\Delta Q = \text{const } j_c B \Delta x$$

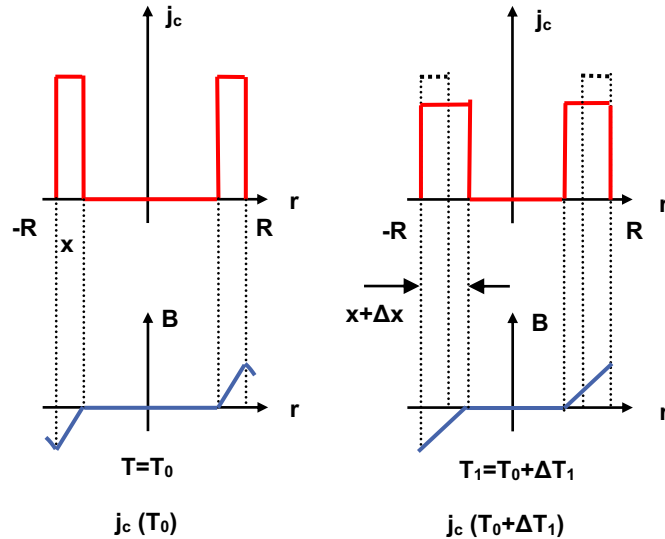
ΔQ results in a temperature increase ΔT_2 (adiabatisch)

$$\Delta T_2 = \frac{\Delta Q}{\rho_{SL} c_{SL}}$$

Thermal stability only if

$$\frac{\Delta T_2}{\Delta T_1} < 1 \quad \Rightarrow \quad x < j_c^{-1} \sqrt{\frac{3 \rho_{SL} c_{SL}}{\mu_0}} T_0$$

Intrinsic stability



Thermal stability only if

$$\frac{\Delta T_2}{\Delta T_1} < 1 \quad \Rightarrow \quad x < j_c^{-1} \sqrt{\frac{3 \rho_{SL} c_{SL}}{\mu_0}} T_0$$

Example for NbTi

$$j_c = 3 \cdot 10^9 \frac{A}{m^2} \text{ bei ca. } 4T$$

$$\rho_{SL} = 6,2 \cdot 10^3 \frac{kg}{m^3}$$

$$c_{SL} = 0,89 \frac{Ws}{kg K}$$

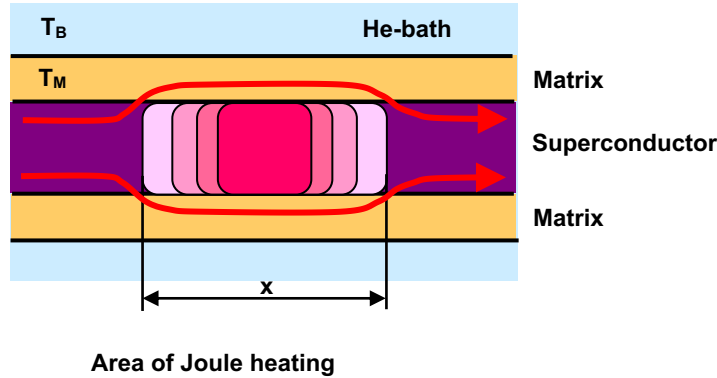
$$T_0 = 4 K$$

$$x < 76 \mu m$$

Stability of the sc state in sc wires

- Thermal balance
- Current sharing
- Intrinsic filament stability
- **Electric stability with cooling**

Stabilization by cooling

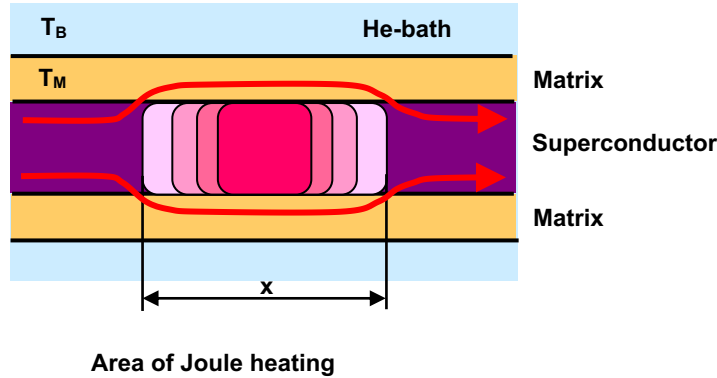


Assumptions:

- Within the normal conducting zone current flows only through matrix
- No contact resistance between SC and matrix
- No axial heat transfer
- Stationary conditions $d/dt=0$
- No external heat sources

$$c_p(T)\rho V \frac{\partial T}{\partial t} = Q_{ext} + Q_{Joule} + \frac{\partial}{\partial x} \left(\lambda(T) \frac{\partial T}{\partial x} \right) A_q - \alpha(T) A_o (T - T_{Kühl})$$

Stabilization by cooling



Resistive heating within the matrix

$$Q_{Joule} = \frac{I^2 \rho_M x}{A_M}$$

Heat transfer to cooling bath

$$Q_B = \alpha_{MB} (T_M - T_B) A_{MB} \text{ mit } A_{MB} = u_{MB} x$$

No extension of the normal conducting area if

$$Q_{Joule} < Q_B \text{ und } T_M < T_C$$

Therefore

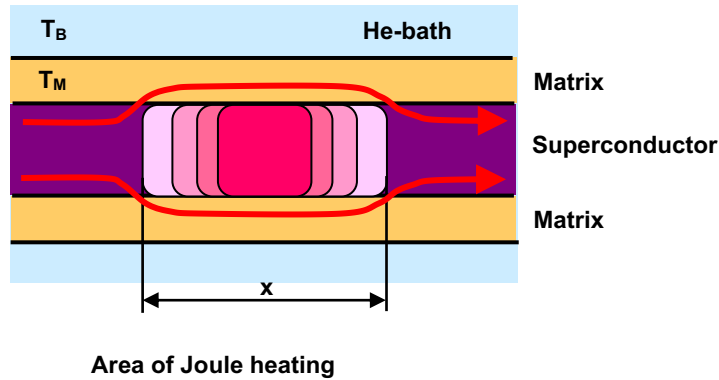
$$\alpha_{st} = \frac{I^2 \rho_M}{A_M u_M \alpha_{MB} (T_C(B, j_{SL}) - T_B)} < 1$$

For stable temperature $Q_M = Q_B$

$$\frac{I^2 \rho_M}{A_M} = \alpha_{MB} (T_C(B, j_{SL}) - T_B) u_{MB}$$

Stekly, Zar, Stable superconducting coils, IEEE Transactions on Nuclear Science, 1965

Stabilization by cooling



↪ Evaporation

For stable temperature $Q_M = Q_B$

$$\frac{I^2 \rho_M}{A_M} = \alpha_{MB} (T_C(B, j_{SL}) - T_B) u_{MB}$$

Example for
NbTi wire without matrix

$$I = 50 \text{ A bei } 5 \text{ T}$$

$$d = 0,25 \text{ mm}$$

$$\rho_{SL} = 1 \cdot 10^{-4} \Omega \text{ cm}$$

$$\alpha_{MB} = 0,7 \frac{\text{W}}{\text{cm}^2 \text{ K}}$$

$$\Delta T = T_{SL} - T_B = 9200 \text{ K}$$

Example for
NbTi wire with matrix

$$I = 50 \text{ A bei } 5 \text{ T}$$

$$d_{CU} = 0,125 \text{ mm}$$

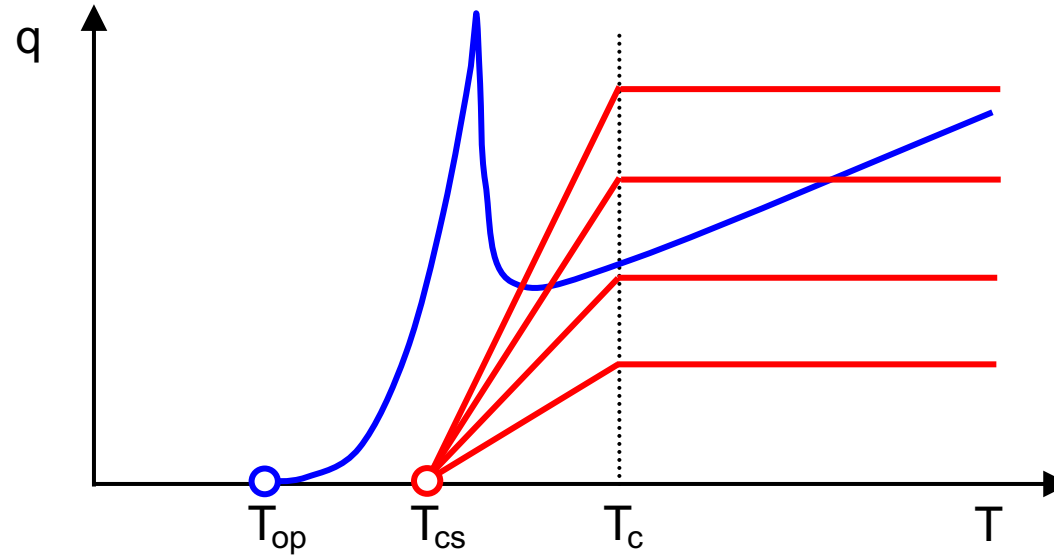
$$\rho_{Cu} = 1 \cdot 10^{-8} \Omega \text{ cm (4 K)}$$

$$\alpha_{MB} = 0,7 \frac{\text{W}}{\text{cm}^2 \text{ K}}$$

$$\Delta T = T_{SL} - T_B = 0,4 \text{ K}$$

Stabilization by bath cooling

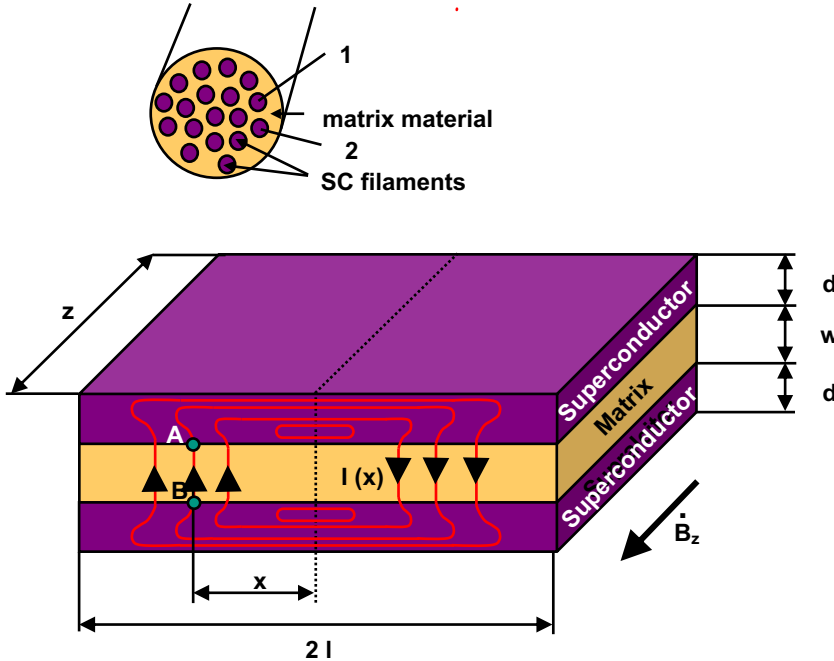
- Cryostability



Losses in sc wires due to non stationary conditions

- Filament coupling
- AC-losses (hysteretic losses, eddy current losses)

Filament Coupling in Multifilamentary Wires



Critical length l_c

$$l_c = \sqrt{2 \rho_M j_c \frac{d}{\dot{B}_z}}$$

Current between SC layers

$$I = \int_0^l j(x) z dx$$

from

$$\oint E ds = - \frac{d\phi}{dt}$$

results

$$U_{AB} = -x w \dot{B}_z = j(x) w \rho_M$$

and

$$j(x) = - \frac{x \dot{B}_z}{\rho_M}$$

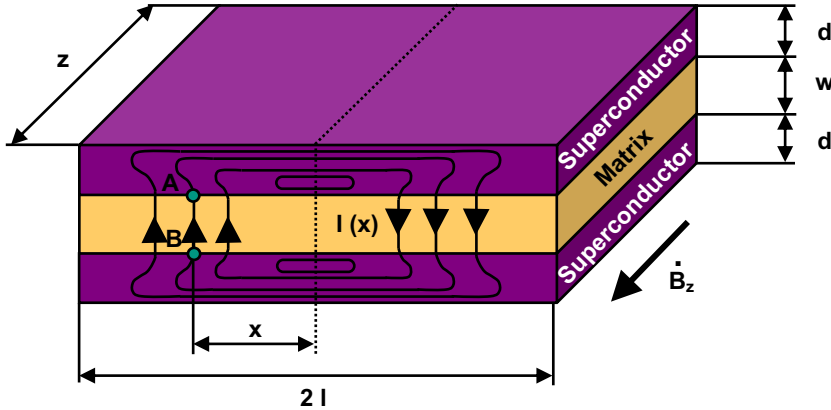
Therefore

$$I = - \frac{\dot{B}_z l^2}{2 \rho_M} z$$

Maximal current at $l_{\max} = l_c$

$$I_{\max} = I_C = j_c d z$$

Filament Coupling in Multifilamentary Wires



$$l_c = \sqrt{2 \rho_M j_c \frac{d}{\dot{B}_z}} \quad \text{Critical length } l_c$$

example for NbTi-conductor
in copper matrix

$$j_c = 2 \cdot 10^9 \frac{A}{m^2}$$

$$d = 50 \mu m$$

$$\dot{B} = 0,1 \frac{T}{s}$$

$$\rho_M = 4 \cdot 10^{-10} \Omega m$$

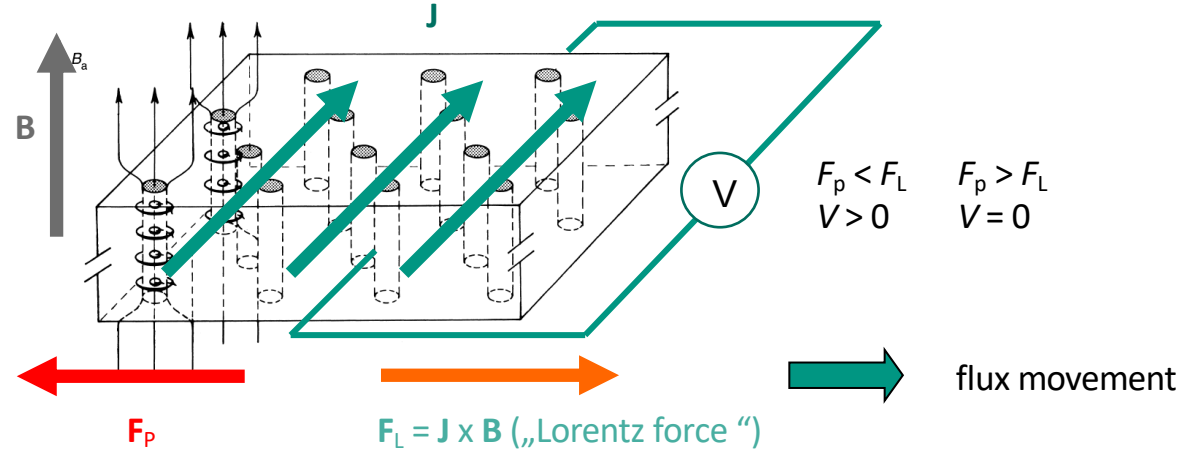
$$l_c = 2,8 cm$$

Losses in sc wires due to non stationary conditions

- Filament coupling
- AC-losses (hysteretic losses, eddy current losses)

AC-losses in Superconductors

Reminder: current transport in real type II Superconductors



Electrical current density j_T results in a Lorentz force F_L on each flux line

For $J > J_c$ ($= F_L > F_p$) flux lines move and dissipate energy

For AC-currents flux penetration changes periodically due to self field oscillation and shows hysteretic behaviour → “hysteresis losses” (see last tutorial)

AC-losses in Superconductors

Self field hysteresis losses

Assumption:

- Bean Modell (homogenous, $+j_c/-j_c$, $J_c = \text{const}$)

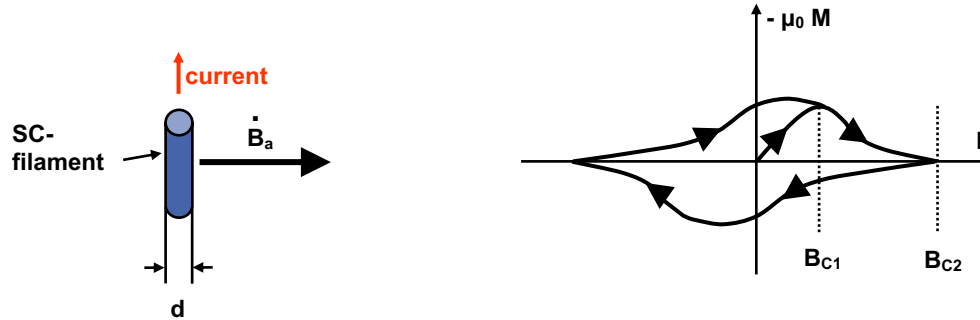
Hysteresis losses in J/m/period

elliptical conductor	$P = \frac{I_c^2 \mu_0}{\pi} \left\{ (1-i) \ln(1-i) + i - \frac{1}{2} i^2 \right\}$	$i = \frac{I_0}{I_c}$
rectangle conductor	$P = \frac{I_c^2 \mu_0}{\pi} \left\{ (1-i) \ln(1-i) + (1+i) \ln(1+i) - i^2 \right\}$	$i = \frac{I_0}{I_c}$

W. T. Norris. Calculation of hysteresis losses in hard superconductors carrying ac: isolated conductors and edges of thin sheets, J. Phys. D, vol. 3, 489-507, 1970

AC-losses in Superconductors

External field variations – hysteresis losses



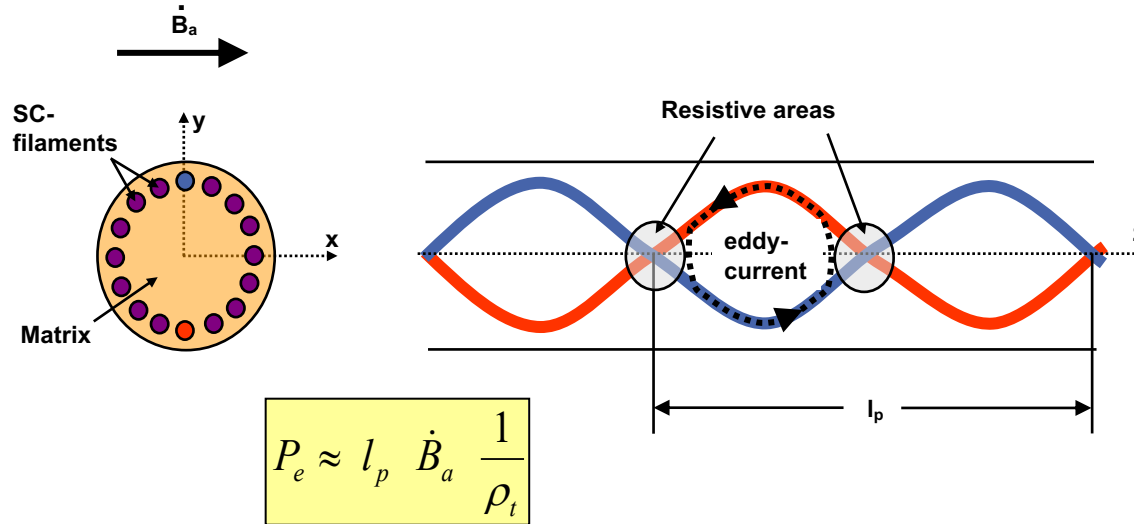
$$P_h \approx d \dot{B}_a j_c V$$

Reduction of hysteresis losses by

- thinner filaments (even thinner as necessary for stabilization)

AC-losses in Superconductors

External field variations – eddy current losses

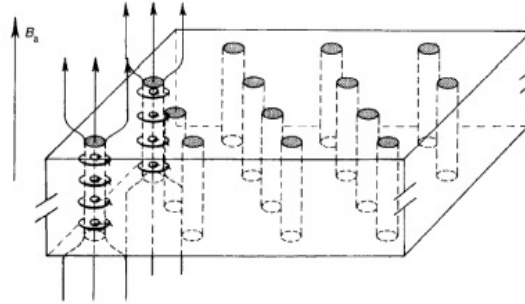
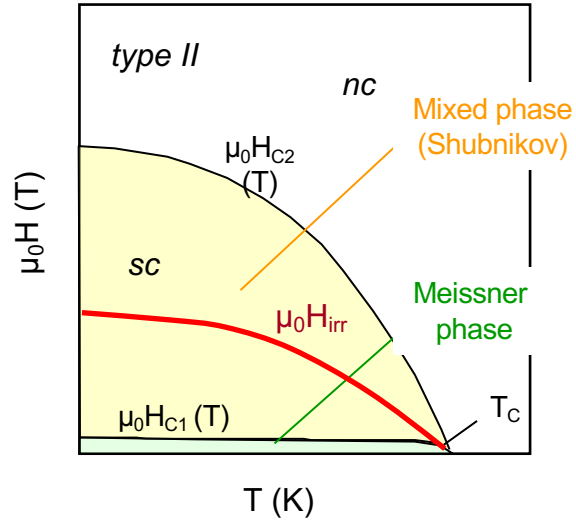


Reduction of eddy current losses by

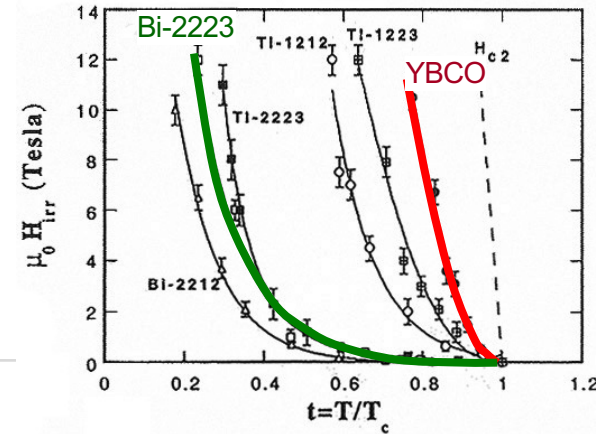
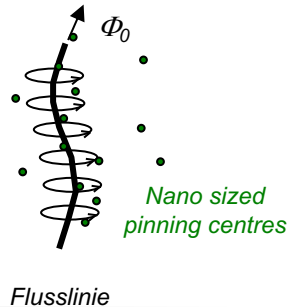
- matrix with barriers to realize high transversal specific resistance ρ_t
- small twist length l_p

Irreversibility field of HTS

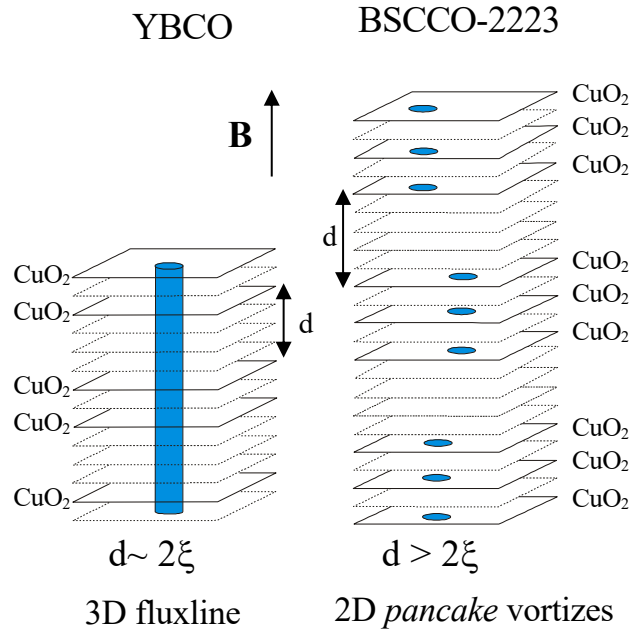
$B(T)$ phase diagram



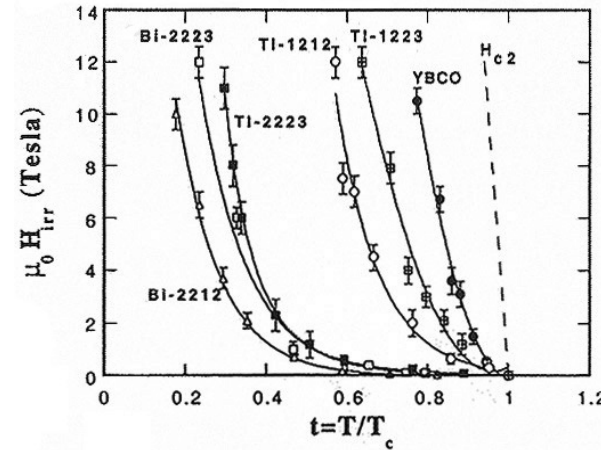
Irreversibility field limits applicability



Irreversibility field of HTS



Irreversibility fields



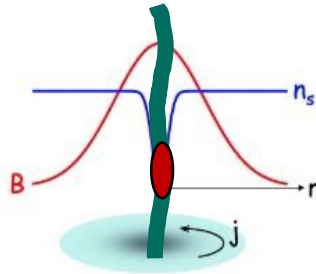
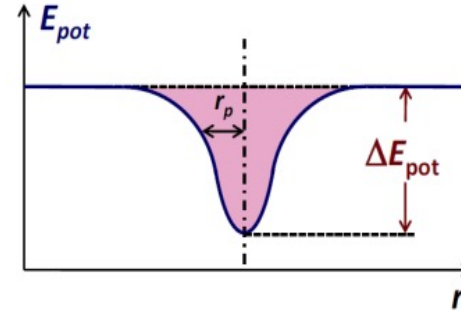
Very small coherence length in cuprates (few nm) + $\xi_c < \xi_{ab}$!

Low anisotropy vs. high T_c
at and below 77K YBCO has highest H_{irr} !

Pinning aspects

Pinning means always the preferred presence of flux lines at positions, where their local potential energy is reduced

$$F_p = -\frac{\partial E_{\text{pot}}(r)}{\partial r} \approx -\frac{\Delta E_{\text{pot}}}{r_p}$$



E.g. core- pinning:

Flux line pinned by normal conducting precipitates

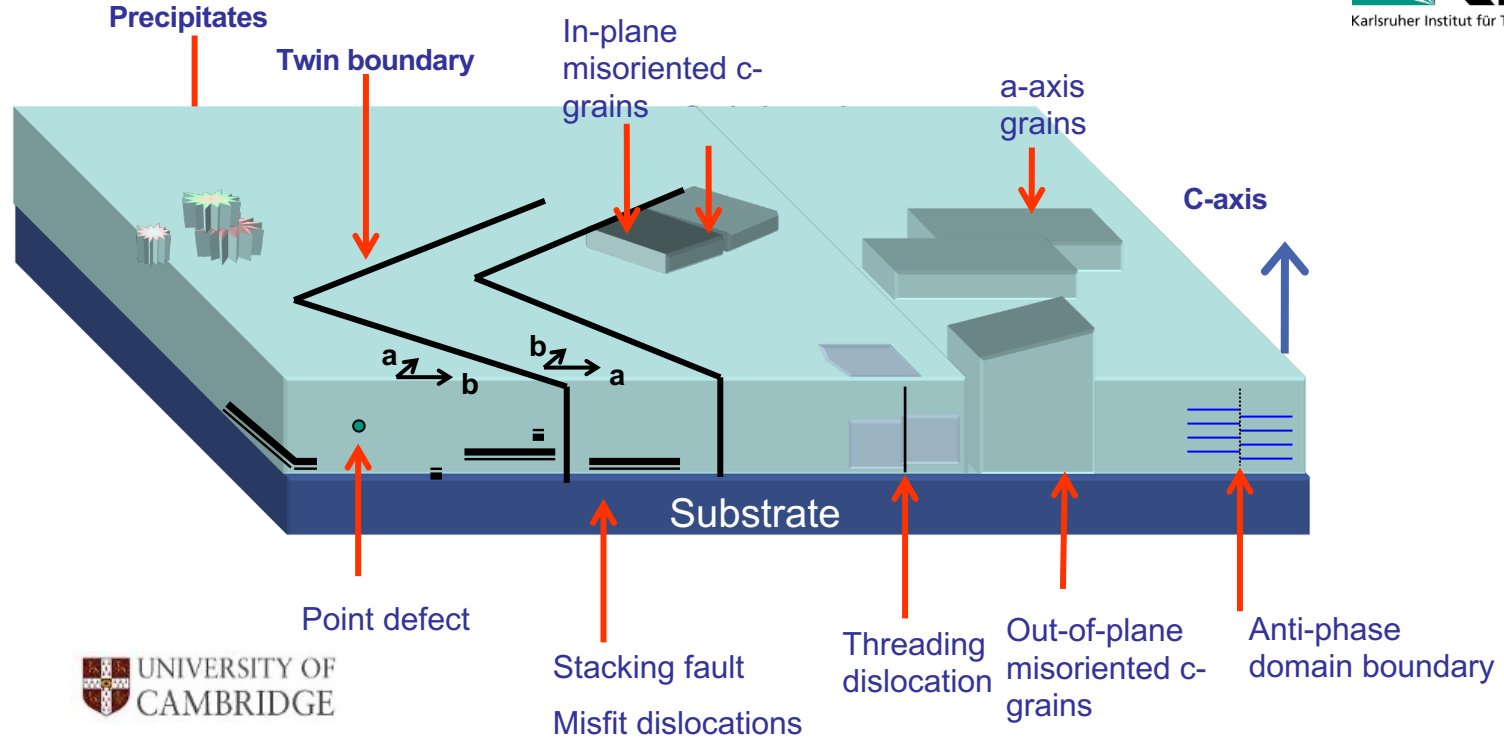
→ less Cooper pairs need to be broken

→ reduction of condensation energy penalty at flux line core

Maximum pinning force, if $\varnothing \sim \xi \rightarrow$ nanotechnology

Other ways to pin FL: Magnetic interaction, strain/stress, T_c oder κ fluctuations, collective pinning of flux line bundles, intrinsic pinning of layered crystal structure ...

Native Defects in HTS Films

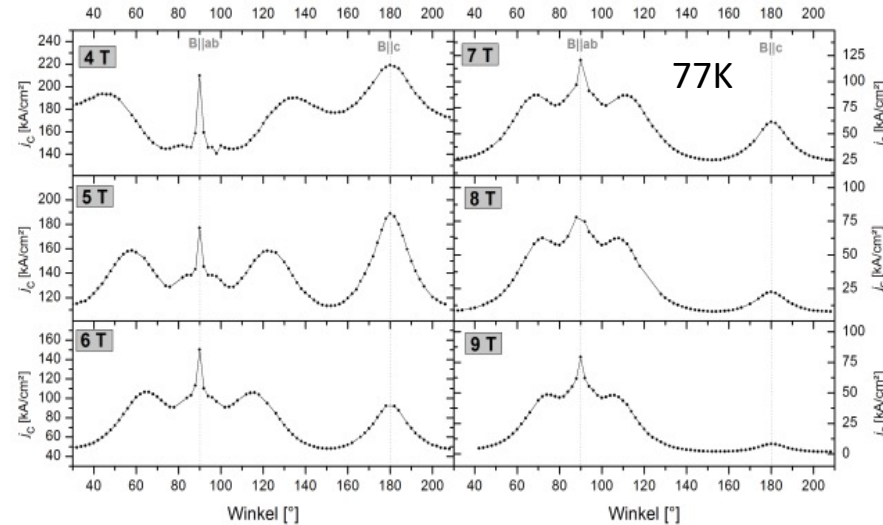
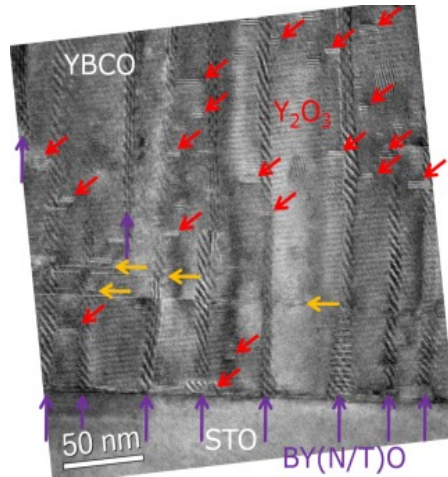


Isotropic: nanodots, strain, disorder, point defect,...

Anisotropic: dislocations (thread., misfit, partial), TB, sGB, strain, SF, APB, buckling, intrinsic, nanopillars,...

Complex pinning landscapes in HTSC

Pinning Engineering through incorporation of various defect structures (here $\text{Ba}_2\text{YNbO}_6/\text{Ba}_2\text{YTaO}_6$ -Nanocolumns, Y_2O_3 -Nanoparticles, stacking defects) within the superconducting matrix (YBCO)



→ Very complex and variable J_c angular dependence in HTSC